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# **Dependence of the Seebeck Coefficient on Specific and Universal Electrical Conductivities of Bi2Sr2Co1.8O<sup>y</sup> Thermoelectric Doped with Strontium Borate and Graphene**

# **Irakli Nakhutsrishvili a\*, Giorgi Kakhniashvili <sup>a</sup> , Iamze Kvartskhava <sup>a</sup> and Akaki Gigineishvili <sup>b</sup>**

*a Institute of Cybernetics of Georgian Technical University, Tbilisi, Georgia. <sup>b</sup> Department of Engineering Physics, Georgian Technical University, Tbilisi, Georgia.*

# *Authors' contributions*

*This work was carried out in collaboration among all authors. All authors read and approved the final manuscript.*

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# **ABSTRACT**

The Seebeck effect has found its application in many areas of human activity. This effect is applicable in the thermoelectric generators for converting waste heat into electrical energy.  $Bi<sub>2</sub>Sr<sub>2</sub>Co<sub>1.8</sub>O<sub>y</sub>$  ceramics is a promising thermoelectric material. Incorporation of suitable dopants into the Bi<sub>2</sub>Sr<sub>2</sub>C<sub>01.8</sub>O<sub>y</sub> host matrix significantly increases the thermoelectric performance of this system. This paper considers the dependence of the Seebeck coefficient on the specific and universal electrical conductivities of Bi<sub>2</sub>Sr<sub>2</sub>Co<sub>1.8</sub>O<sub>y</sub> thermoelectric doped with strontium borate — Sr(BO<sub>2</sub>)<sub>2</sub> and graphene. It is shown that the dependences of the Seebeck coefficient on the electrical conductivity in the doped compositions are rectilinear for individual samples. The dependences of the Seebeck coefficient on the universal electrical conductivity exhibit a power-law character, but their form is

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*<sup>\*</sup>Corresponding author: Email: iraklinakhutsrishvili52@gmail.com;*

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practically independent of the dopant concentrations. The temperature dependences of the electronic quality factor ( $Be$ ) are also investigated. An increase of  $Be$  with temperature indicates the presence of effects of additional scattering and band convergence.

*Keywords: Bi2Sr2Co1.8O<sup>y</sup> thermoelectric; doping; electrical conductivity.*

# **1. INTRODUCTION**

Because of their ability to directly convert waste heat into electrical power, thermoelectric materials have attracted considerable interest as a source of environmentally friendly energy to address the energy crisis and ecological issues. Layered cobaltites are promising materials for high-temperature thermoelectric generators [1,2]. Doping method is widely used in order to enhance their functional efficiency, in particular, increase the power factor  $PF = \sigma S^2$ , where  $\sigma$  is the electrical conductivity, S is the Seebeck coefficient [3-5]. Based on previously reported data, the dependences of the Seebeck coefficient on the specific and universal electrical conductivities in  $Sr(BO<sub>2</sub>)<sub>2</sub>$  [6] and graphenedoped [7] Bi<sub>2</sub>Sr<sub>2</sub>Co<sub>1.8</sub>O<sub>y</sub> thermoelectrics were investigated in this paper.

#### **2. METHODOLOGY**

In the theory of semiconductors, in the relevant sections, the dependence of the Seebeck coefficient on the characteristics of charge carriers is defined as [8-10]:

$$
S = \frac{8\pi^2 k_B^2}{3q h^2} m^* T \left(\frac{\pi}{3n}\right)^{2/3},
$$
 (1)

where n is the concentration of charge carriers, m\* is the effective mass, q is the elementary charge,  $T$  is absolute temperature,  $k_B$  and h are the Boltzmann and Planck constants, respectively. Taking into account the expression for electrical conductivity σ=nqμ (μ - mobility) and

the values of universal constants, formula (1) will take the form:

$$
S = 2.17 \cdot 10^{-16} m^* \mu^{2/3} T \sigma^{-2/3}.
$$
 (2)

The concept of universal electrical conductivity is also introduced:

$$
\sigma' = \frac{\sigma}{B_E} \left(\frac{q}{k_B}\right)^2,\tag{3}
$$

where  $B_E = \sigma S^2 / B_S$  is the electronic quality factor, and B<sup>S</sup> is a dimensionless quantity (scaled power factor) depending on the Seebeck coefficient [11]. Using Eqs.(2) and (3):

$$
S = 5.7 \cdot 10^{-11} m^* \mu^{2/3} T B_E^{-2/3} (\sigma')^{-2/3} . \tag{4}
$$

In this paper, we consider the dependences of the Seebeck coefficient on the specific and universal electrical conductivities in  $Bi<sub>2</sub>Sr<sub>2</sub>Co<sub>1.8</sub>O<sub>v</sub>$ thermoelectric doped with  $Sr(BO<sub>2</sub>)<sub>2</sub>$  and graphene.

#### **3. RESULTS AND DISCUSSION**

It should be noted that the comparative narrowness of the range of the Seebeck coefficient change ((1.05-1.78 )10-4V∙K-1 ) makes it possible to consider the dependence of the Seebeck coefficient on the electrical conductivity for our samples in a simpler way. The study of the relationship between the power factor and the Seebeck coefficient showed that for all doped samples the dependences  $\sigma S^2 - S$  are rectilinear (Fig. 1):



**Fig. 1. Dependences of the power factor on the Seebeck coefficient: Bi2**− [( ) ]. − **(o) x=0.075, (∆) x=0.1, (□) x=0.15; Bi2Sr2Co1.8Oy+ xGr** − **(●) x=0.35, (▲) x=0.7, (■) x=1.15**

x	k, Sim $(K·m)$ <sup>1</sup> V	$10^5$ b, W $\cdot$ K $^{-2}$ ·m $^{-1}$
0.075	0.22	$-1.3$
0.1	0.5	$-2.35$
0.15	0.4	$-2.2$
0.35	0.39	$-2.55$
0.7	0.45	$-2.8$
1.15	0.3	$-2$

**Тable 1. Values of constants in Eq.(5) for different x (=0.075-0.15: Sr(BO2)2, =0.35-1.15: Gr)**



**Fig. 2. Implicit dependences**  $S - \sigma$ :  $Bi_2Sr_{2-x}[Sr(BO_2)_2]_xCo_{1.8}O_y -$  (o) x=0.075, (∆) x=0.1, (□) **x=0.15; Bi2Sr2Co1.8Oy+ xGr** − **(●) x=0.35, (▲) x=0.7, (■) x=1.15**

$$
\sigma S^2 = kS + b,\tag{5}
$$

$$
(5)
$$

where k is the slope of the lines, b is the ordinate of the point of intersection of these lines with the σ axis during their extrapolation (the values of k and b are given in the table).

Eq.(5) can be rewritten as:

$$
=\frac{kS+b}{S^2}=\frac{k}{S}+\frac{b}{S^2}
$$
 (6)

Fig.2 shows the dependence S– σ in the implicit form. Graph of Eq.  $(6)$  is a curve of the  $3<sup>rd</sup>$  order, but due to the small range of change in S, we have segments in the form of almost straight lines. For comparison, we plotted these dependences for larger ranges of S change (up to  $(2.5\n-5) \cdot 10^{-4}V \cdot K^{-1}$  for other thermoelectrics [12,13]. A deviation from straightness was observed, which follows from the above formulas.

It can be seen from Fig. 2 that for most samples, an increase in S leads to a decrease in σ (and vice versa), which also follows from the above formulas. Since σ and S depend on temperature, an increase in the latter leads to an increase in the power factor for all samples. A study of the dependence of the power factor on σ and S separately showed that  $\sigma S^2$  decreases with increasing σ and increases as S increases (according to data of [6] and [7]). Since  $σS<sup>2</sup>$ depends on S more than on σ, this results is an increase of the power factor.

Taking  $B_E = \sigma S^2/B_s$  into account, formula (3) will take the form:

$$
\sigma' = \left(\frac{q}{k_B}\right)^2 \frac{B_S}{S^2} = \left(\frac{q}{k_B}\right)^2 \frac{\sigma}{B_E} \cong 1.347 \times 10^8 \frac{B_S}{S^2} \tag{7}
$$

or

$$
S = 1.16 * 104BS1/2(\sigma')-1/2.
$$
 (8)

Dependences S–σ' according to the formula (8) for the studied samples are shown in Fig. 3. It can be seen that the experimental points here also form almost a single set, regardless of the concentration of the dopants (i.e.  $B_E$  scales electrical conductivity). Their combination can be described by a single empirical expression  $S \cong 6.79 \cdot 10^4 (\sigma')^{-0.526} - 1.5 \cdot 10^{-5}$ . (Obviously, the dependence  $\sigma' - B_S/S^2$ , constructed according to formula (7), will have the same form for any thermoelectric  $-$  the form of a straight line with a slope of ≅1.347∗10<sup>8</sup> SimW-1K<sup>2</sup> .)

In addition to the fact that electronic quality factor BE scales thermoelectric parameters (electrical conductivity, power factor), it is also should be noted that  $B_E$  does not depend on temperature for an ideal material. A deviation from this indicates the presence of additional effects.

To determine BE, we first calculated the values of  $B<sub>S</sub>$  using the formula [11]:

$$
B_{S} = \frac{\left(\frac{qS}{kg}\right)^{2} e^{-2\frac{qS}{kg}}}{1+e^{-5\left(\frac{qS}{kg}-1\right)}} + \frac{\frac{\pi^{2}qS}{3kg}}{1+e^{\frac{q(S-1)}{kg}}}
$$
(9)

The temperature dependences of parameters obtained by the equation (9) and  $B_E = \sigma S^2/B_S$  are shown in Fig. 4. The values of  $B<sub>s</sub>$  change slightly for both types of samples; values of  $B<sub>E</sub>$  are practically constant at first (a sign of ideal case), and then increase. An increase of B<sup>E</sup> with temperature indicates the presence of such effects as additional scattering and band convergence [11].



**Fig. 3. S–σ' dependences: (o)** −  $Bi_2Sr_{2-x}[Sr(BO_2)_2]_xCo_{1.8}O_y$ , (●) −  $Bi_2Sr_2Co_{1.8}O_y$ + xGr



Fig. 4. Typical temperature dependences of B<sub>E</sub> and B<sub>S</sub>: o, ∆ -  $\rm Bi_2Sr_{1.925}[Sr(BO_2)_2]_{0.075}Co_{1.8}O_y$ ; •,  $\triangle$  -  $\text{Bi}_2\text{Sr}_2\text{Co}_{1.8}\text{O}_v$  + 0.35Gr

# **4. CONCLUSION**

Thus, it can be stated that the dependences of the Seebeck coefficient on the electrical conductivity in the  $Bi_2Sr_2Co_{1.8}O_y$  thermoelectric ceramics doped with the strontium borate and graphene are rectilinear for individual samples. This provides a simple way to calculate the  $S - \sigma$ dependence without using formula (1). The dependences of the Seebeck coefficient on the universal electrical conductivity exhibit a powerlaw character, practically do not depend on the concentration of dopants (i.e. B<sup>E</sup> scales electrical conductivity), and can be described by a single empirical expression  $S \cong 6.79 \cdot 10^4 (\sigma')^{-0.526} - 1.5$ 10−5 . For all the samples, the values of the electronic quality factor were calculated and their temperature dependences were plotted. Values of  $B<sub>E</sub>$  are practically constant up to 300 °C for  $Bi_2Sr_{1.925}[Sr(BO_2)_2]_{0.075}Co_{1.8}O_y$  and 500 °C for  $Bi_2Sr_2Co_{1.8}O_v + 0.35$  wt % Gr, a sign of ideal case, and then increase. An increase of  $B<sub>E</sub>$  with temperature indicates the presence of additional effects (additional scattering and band convergence).

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# **COMPETING INTERESTS**

Authors have declared that no competing interests exist.

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# **APPENDIX**

 $Bi_2Sr_2\times [Sr(BO_2)_2]_xCo_{1.8}O_y(x = 0.075, 0.1 \text{ and } 0.15)$  and  $Bi_2Sr_2Co_{1.8}O_y + x$  wt. % Gr (x=0.35, 0.7 and 1.15) ceramic samples were prepared by standard solid-state reaction method from reagent-grade powders of bismuth oxide ( $Bi_2O_3$ ), strontium carbonate (SrCO<sub>3</sub>), cobalt oxide (Co<sub>3</sub>O<sub>4</sub>) and graphene nanopowder ( purity: 99.2%, average flake thickness: 12 nm (30–50 monolayers), average particle (lateral) size: 4.5 µm).

(a)  $Bi_2Sr_2\times[Sr(BO_2)_2] \times Co_{1.8}O_v$ : mixtures of initial powders were mixed and subjected to heat treatment at 750–830°C for 30 hours with intermediate grinding in an agate mortar at 830°C for 25 hours. Then the powders were pressed into pellets with a diameter of 15 mm at a hydrostatic pressure of 200 MPa. Finally, the pellets were annealed at 830°C. Cooling to room temperature was carried out for 15 h in the furnace.

(b)  $Bi<sub>2</sub>Sr<sub>2</sub>Co<sub>1.8</sub>O<sub>v</sub> + xGr.$  The mixtures of powders were homogenized in a planetary mill (Fritsch Pulverisette 7 Premium line) for 1 h at a rotating speed of 120 rpm. After homogenization, the powders were calcined at 770–815°C for 18 hours with intermediate grindings in an agate mortar, then pressed into pellets at a hydrostatic pressure of 220 MPa. Finally, the pellets were sintered at 1103–1108 K in air for 20 h, then cooled to room temperature in the furnace.

The phase purity and microstructure of the prepared materials were examined using X-ray diffraction (Dron–3M, CuKα–radiation) and scanning electron microscopy (VEGA TS5130MM) techniques. The resistivity of the samples as a function of temperature  $p(T)$  was measured by the standard four-probe method. The temperature dependence of the Seebeck coefficient was determined with a homemade setup using a KEITHLEY DMM6500 multimeter.

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